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*Title:* Radiative shocking and acceleration of polycrystalline slabs for investigation of ablative Rayleigh-Taylor instability triggered by ablator microstructure

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# **Radiative shocking and acceleration of polycrystalline slabs for investigation of ablative Rayleigh-Taylor instability triggered by ablator microstructure**

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*Los Alamos National Laboratory*

**44th Annual Meeting  
APS Division of Plasma Physics  
Orlando, Florida  
11-15 November 2002**

**NIF ignition capsules are the most unstable imploding systems ever devised by the national inertial fusion program.**

**Growth rate for ablative Rayleigh-Taylor instability is roughly\***

$$\gamma = \sqrt{k g} - \beta k v_a$$

$g$  = shell acceleration,  $k$  = perturbation wavenumber,  $\beta \sim 2$ ,  
 $v_a$  = velocity of ablation front

**Number of unstable e-foldings for mode  $m$  during acceleration is**

$\gamma t \approx \sqrt{m} - m / \alpha$  where  $m$  = mode number =  $k R_0$ ;  $R_0$  is initial capsule radius

$$\alpha = \frac{\text{shell compression factor} \times \text{initial shell aspect ratio}}{\beta \times \text{fraction of shell ablated}} \sim \frac{4 \times 6.25}{2 \times 0.5} \sim 25$$

Maximum growth  $(\gamma t)_{\max} = \alpha/4 \sim 6.3$  occurs for mode  $m_{\max} = \alpha^2/4 \sim 150$

6.3 e-foldings implies growth factor  $\sim 540$

***Further growth during deceleration leads to total growth factor  $\sim 1000$***

\*H. Takabe et al., *Phys. Fluids* **28**, 3676 (1985); R. Betti et al., *Phys. Plasmas* **3**, 2122 (1996); *Phys. Plasmas* **2**, 3844 (1995).

## **Therefore it is vital to identify and control all perturbation sources that could trigger ablative Rayleigh-Taylor instability**

- **One obvious perturbation “seed” is surface roughness**
  - *Specification for allowable roughness of NIF ignition capsule\* is based on computed perturbation growth, validated by experiments*
- **But what about *internal microstructure* of shell materials?**
  - *Beryllium shells are composed of individual crystalline grains with anisotropic elastic/plastic properties*
  - *Polymer shells are composed of long molecular chains that might “stack like logs” with a preferred orientation*

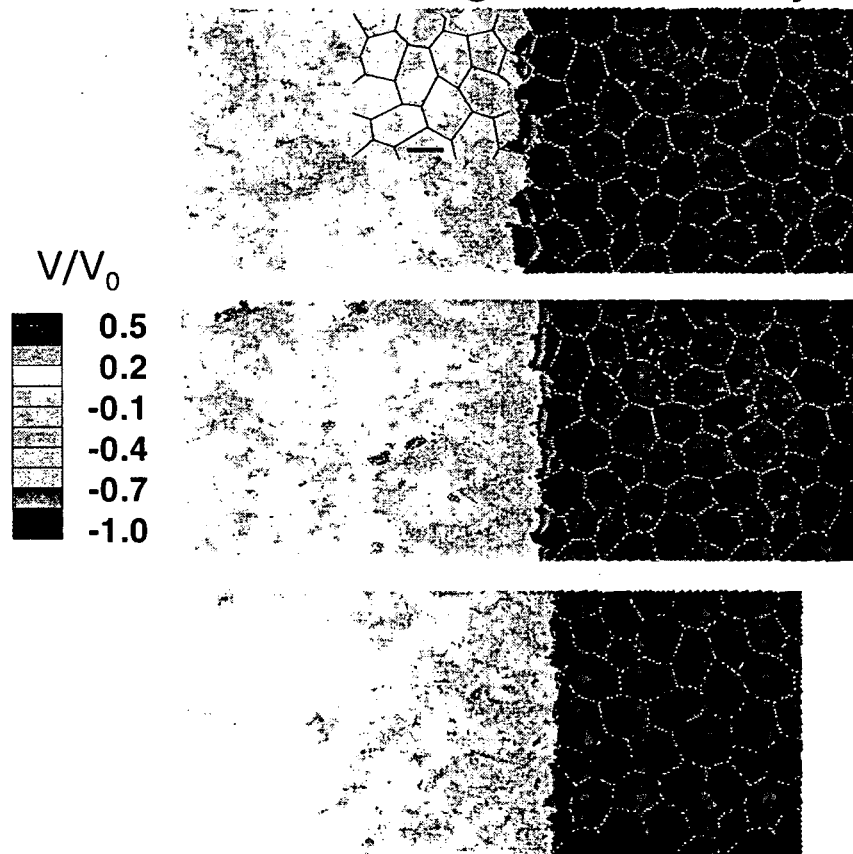
***What happens when shock waves transit anisotropic material?  
What happens when such material is accelerated by radiation drive?  
We need a specification for allowable internal anisotropy.***

\*R.B. Stephens, S.W. Haan, D.C. Wilson, “Characterization Specifications for Baseline Indirect Drive NIF Targets”, internal General Atomics memo

Recent advances allow computation of fluctuating  
velocity field behind shock wave in anisotropic material\*.  
This is likely to seed ablative Rayleigh-Taylor instability.

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Normalized longitudinal velocity fields



Longitudinal Velocity Fluctuations  
in Shock Compression of  
Polycrystalline  $\alpha$ -Iron\*

$V_0 = 150 \text{ m/s}$   $P_s = 5.5 \text{ GPa}$   
—  $10 \mu\text{m}$

$V_0 = 300 \text{ m/s}$   $P_s = 12 \text{ GPa}$

$V_0 = 1000 \text{ m/s}$   $P_s = 45 \text{ GPa}$

\*Y. Horie and K. Yano, "Particle Velocity  
Fluctuations in the Shock Compression  
of Solids", LA-13936-MS, May 2002

## How should we go about developing a specification for ablator microstructure?

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- **Theoretical/computational approach**

- *Develop models, simulate relevant processes numerically*
  - *For example, add radiation to Horie-Yano code, develop 3D Be grain model*
- *This was approach used in developing surface-roughness spec*
- *Experiments are still necessary to validate models and simulations*

- **Empirical approach**

- *Observe behavior of Be/Cu slabs on Omega, Z, early NIF,...*
- *Relate behavior (via computed single-mode growth factors) to **equivalent surface perturbation***
- *Since surface-roughness spec already exists, we thereby reduce microstructure spec to a previously solved problem*
- *Computation is still necessary to determine growth factors, and make diagnostic predictions*

***Either approach must wait for full NIF for final confirmation.  
Similar experiments must be done in either approach.***

## We are proceeding with the empirical approach.

- **Face-on radiography\* of accelerated Be/Cu foils shows growth of unstable perturbations**

- Data consists of time-varying spatial distribution of slab areal density  $\int \rho dz$
- Fourier analysis of areal density distribution gives power spectrum  $S(k,t)$
- Computation gives time-dependent linear growth factor spectrum  $f(k,t)$
- Then “equivalent initial surface perturbation” is

$$P_{eq}(k,t) \equiv S(k,t)/\rho_0 f(k,t)$$

where  $\rho_0$  is initial slab density.

- Perturbation must be linear for this to be valid
- If nonlinear saturation occurs, resort to Haan model\*\*
- Actual initial surface perturbation  $P_0(k,t)$  must be minimized or subtracted
- $P_{eq}(k,t)$  is not a unique attribute of a given microstructure, but depends on radiation drive history, drive spectrum, slab thickness, composition, etc.

- **We will pursue basic theory and modeling at low level**

\*B.A. Remington et al, *Phys. Rev. Lett.* **67**, 3259 (1991)

\*\*S.W. Haan, *Phys. Rev. A* **39**, 5812 (1989)

## First experiments at Omega: Radiography of Be/Cu slabs driven by “hohlraum”

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### Diagnostic efforts include:

- **Hohlraum characterization**
  - *Experiments require long slowly rising drive to maximize growth factor*
  - *Diagnostics: Stepped slab, wedge witness plate, DANTE*
- **Side-on slab radiography**
  - *Slab trajectory, acceleration history*
  - *Verify drive history*
- **Face-on slab radiography**
  - *Time-varying spatial distribution of slab areal density  $\int \rho dz$*

### Experiment geometry and parameters:

- **12 to 15 Omega beams on P6-P7 axis**
  - *~190 eV drive is achievable*
  - *QXI, GXI, FXI gated imaging instruments are available*
- **Slabs**
  - *Type I: Smoothest possible surface, to emphasize volume microstructure*
  - *Type II: Intentional sinusoidal surface perturbation*
    - *Calibrate growth-factor calculations*



## **Initial computational modeling centers on designing slab/foil and radiation drive history for Omega experiments**

- **Goal: choose drive history, slab thickness to give maximum perturbation growth on Omega laser**
  - *Modeling approach: calculate single-mode sinusoidal perturbation growth, for variety of candidate slab thicknesses, drive histories, and spectra*
  - *Use slowly rising pulse to keep slab on low adiabat, minimize preheat*
  - *Set initial shock pressure to maximize velocity anisotropy for ART seeding*
- **Major concern: Is Omega drive strong enough, and microstructure seed large enough, that microstructure-seeded perturbations become visible?**
  - *Line VISAR observations of free-surface velocity for shocked beryllium foils\* give some information about magnitude of velocity fluctuations*
  - *Basic theory could give some guidance here, but needs development*
  - *Meanwhile, simply try to maximize growth factor spectrum with Omega drive*

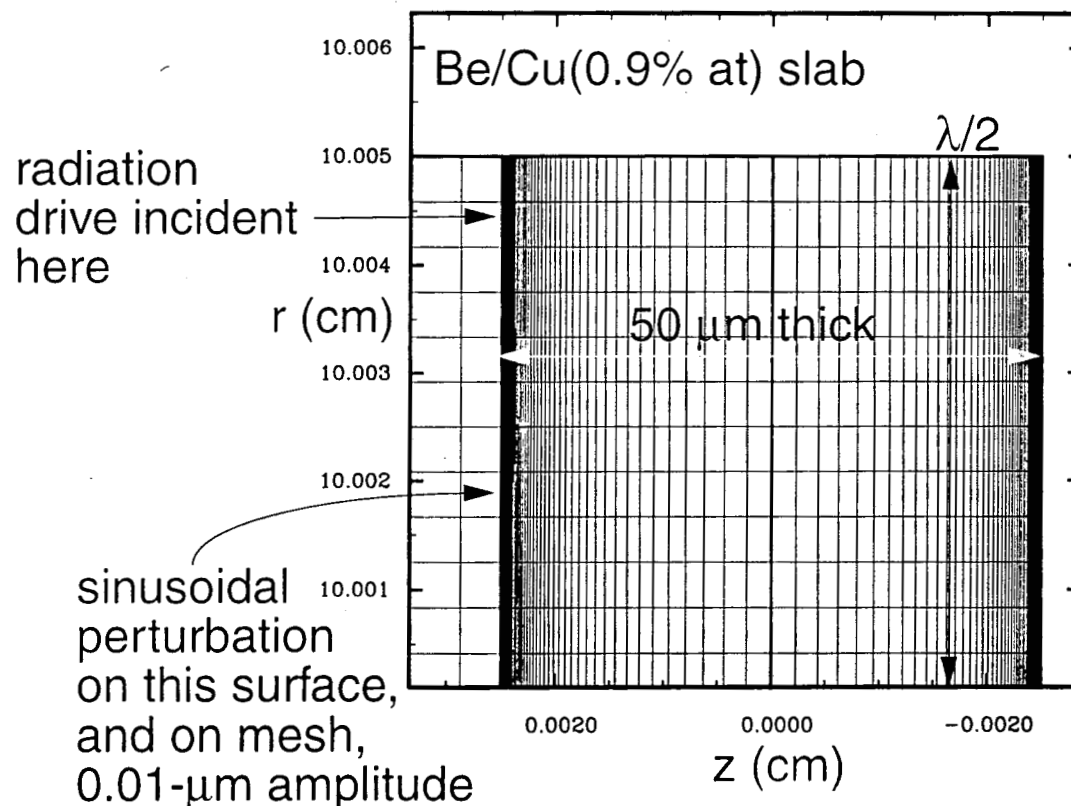
***See poster by Tubbs et al. (this session) for hohlraum designs to produce desired drive histories***

\*D. Swift, LANL report LA-UR-01-6430, 15 November 2001.

## Linear stability modeling consists of calculating growth of single-mode sinusoidal surface perturbation

- Although we model surface perturbation, not microstructural perturbation, we can obtain quantitative measure of slab stability: growth factor spectrum of perturbations

Initial mesh for calculation with perturbation wavelength  $\lambda = 100 \mu\text{m}$



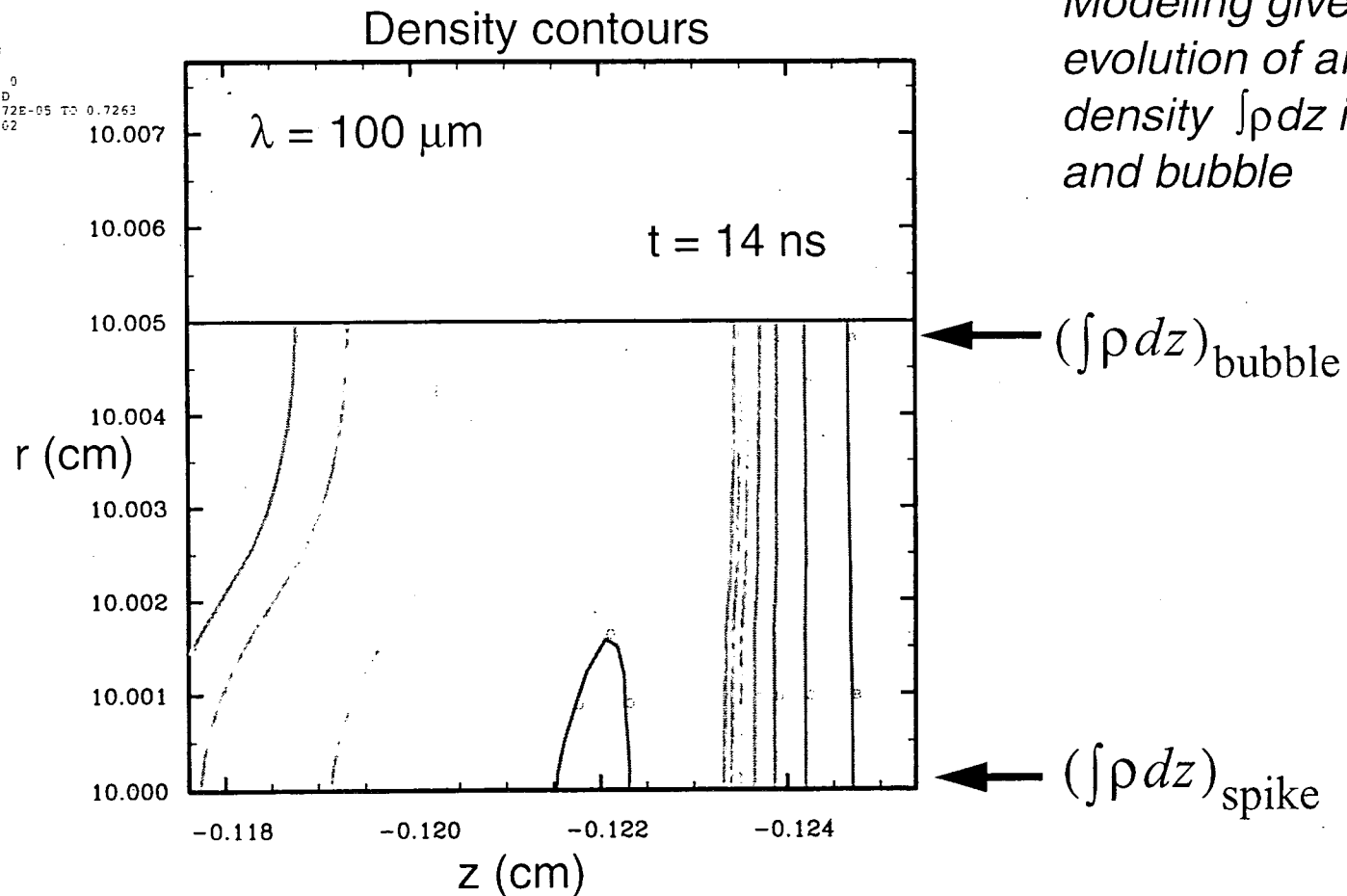
# Linear stability modeling computes development of contrast between peak and valley for small-amplitude single-mode perturbation

```

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* FILE             OF57DE
* TIME             1.400076
* CYCLE            6670
* CODE             VERSION
* PROBLEM NAME OF57
* PROBLEM NUMBER    0
* CONTOURS OF VARIABLE D
* DATA RANGES FROM 4.9972E-05 TO 0.7263
* INTERVAL =       5.0000E-02
* A = 0.0000
* B = 0.0000E+00
* C = 0.0000

```

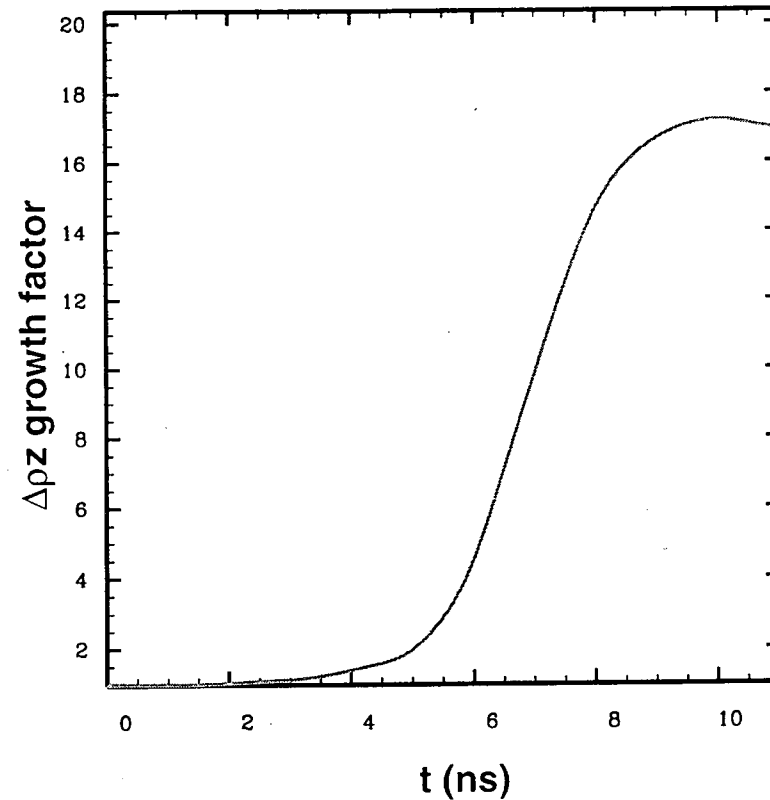
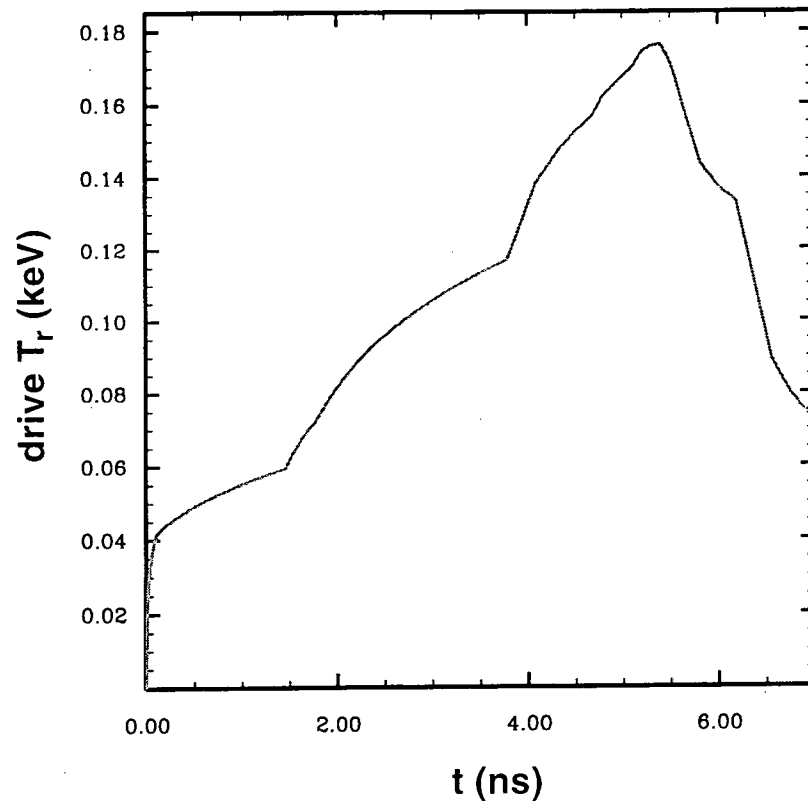
10/30/02 20.13.0



*Modeling gives time evolution of areal density  $\int \rho dz$  in spike and bubble*

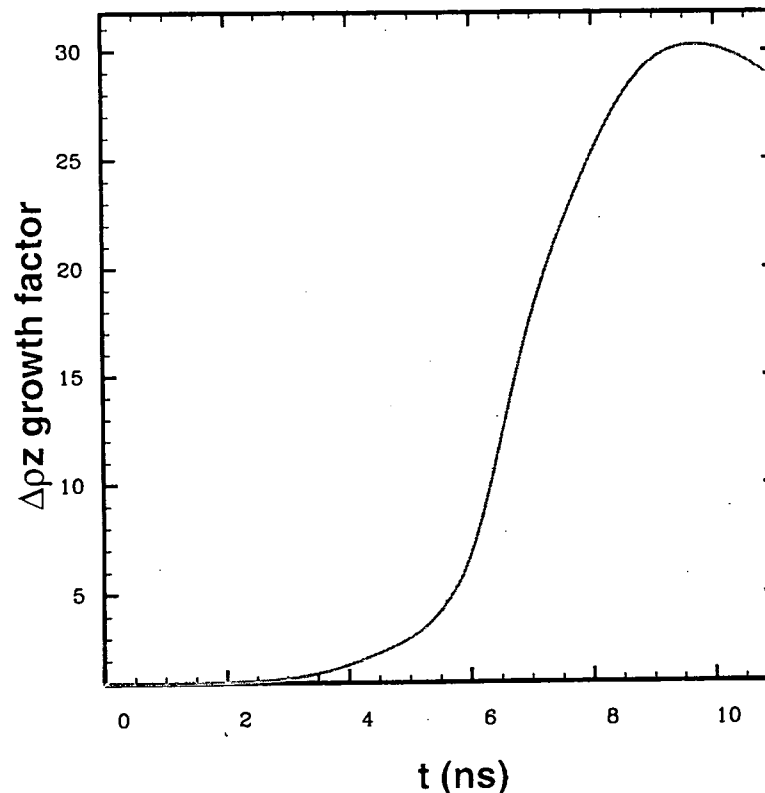
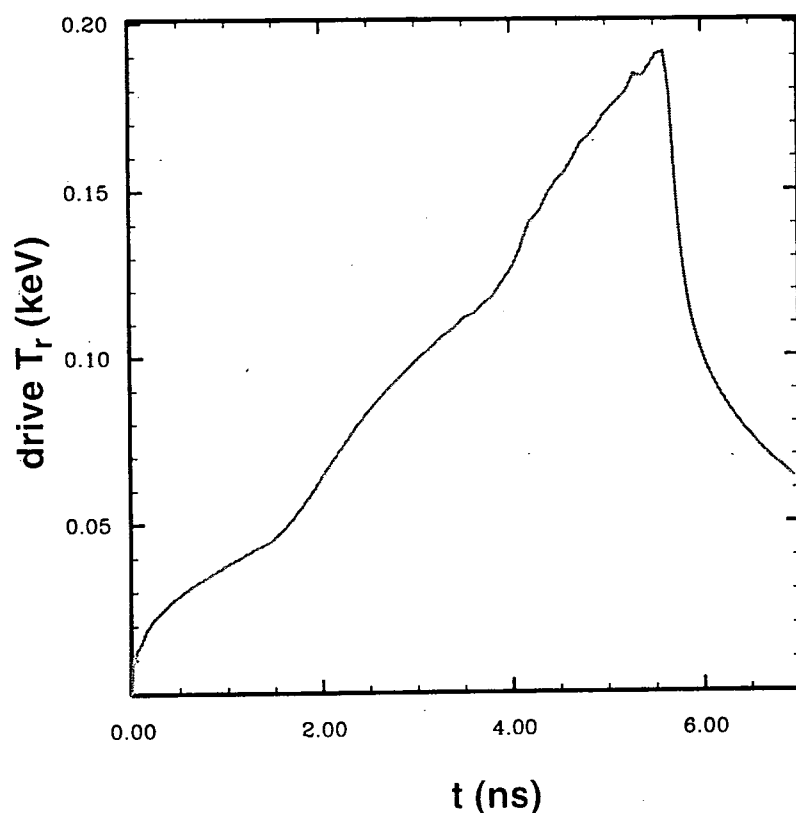
**Trial Omega drive pulse with high-preheat drive spectra gives growth factor ~ 17 for  $\lambda = 100 \mu\text{m}$  and 50- $\mu\text{m}$  foil.**

$$\Delta\rho z \text{ growth factor } (t) \equiv [(\int \rho dz)_{\text{spike}} - (\int \rho dz)_{\text{bubble}}]_t / [(\int \rho dz)_{\text{spike}} - (\int \rho dz)_{\text{bubble}}]_{t=0}$$



of54 (OOmod12)

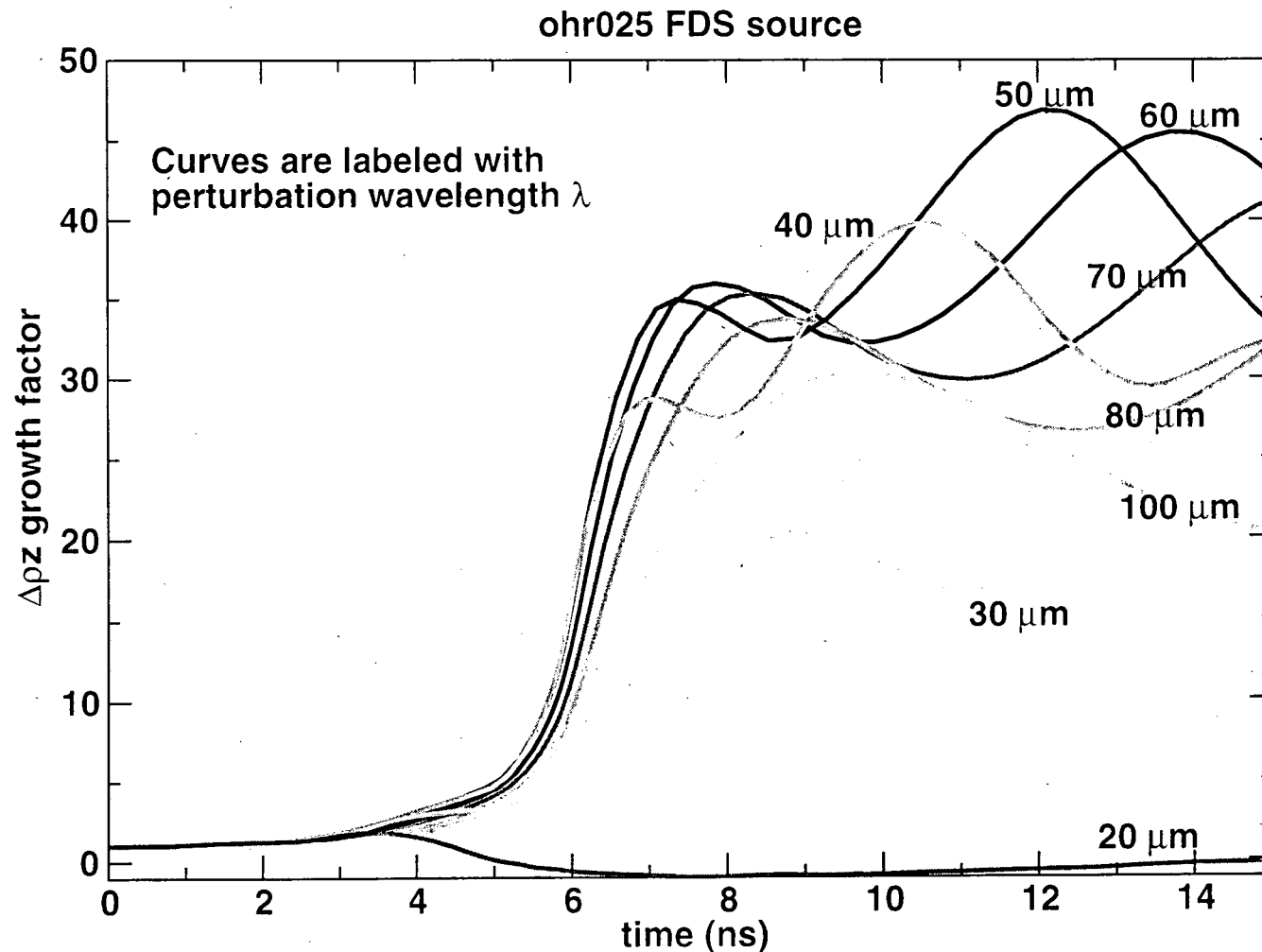
Current proposed Omega drive pulse with realistic calculated spectra gives growth factor  $\sim 30$  for  $\lambda = 100 \mu\text{m}$  and 50- $\mu\text{m}$  foil.



of58,59 (ohr025)

Calculations using current proposed Omega drive pulse with 50- $\mu\text{m}$  foil show greatest growth factor occurs for  $\lambda = 50 \mu\text{m}$ .

### Time-dependent growth factors for 50- $\mu\text{m}$ Be/Cu foil



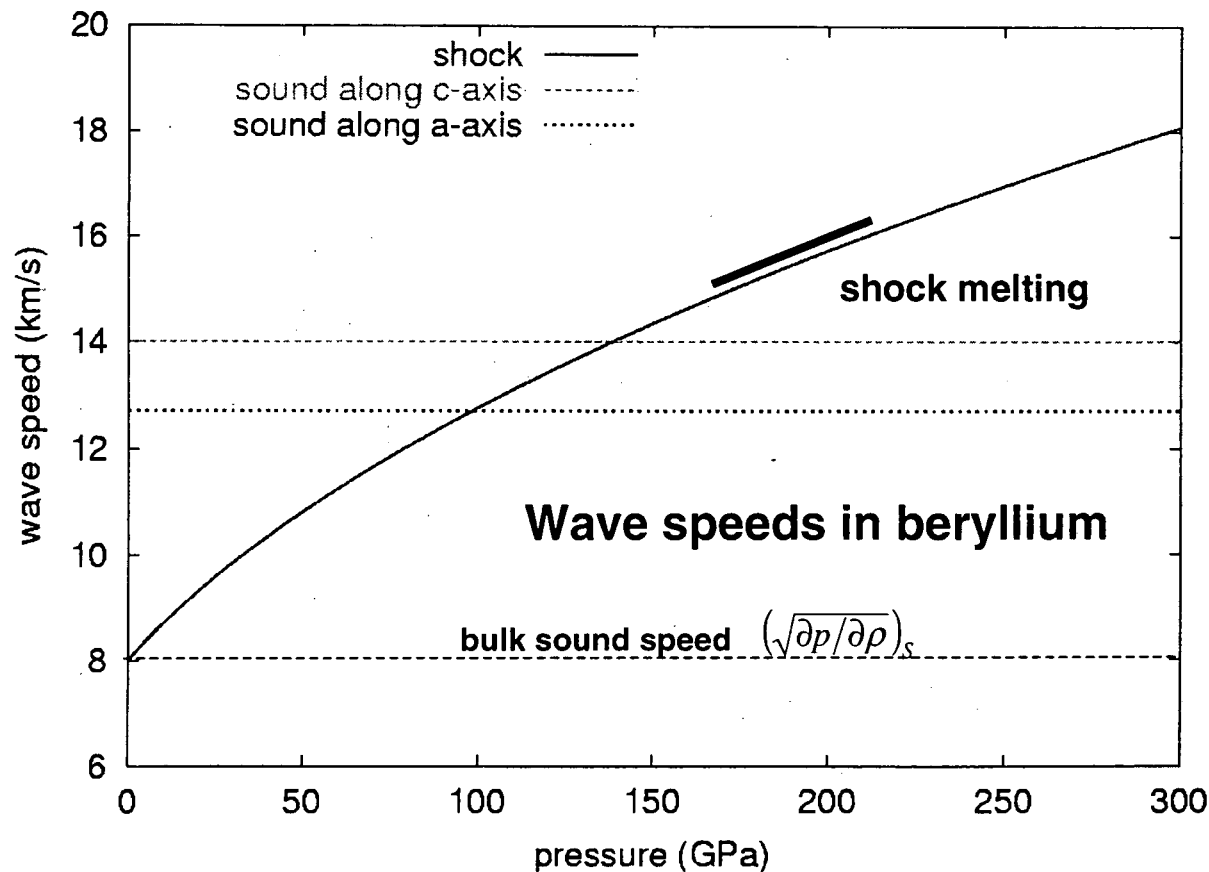
**One goal of experiment is to test influence of shock strength on microstructure-seeded perturbations.**

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- NIF capsules are expected to have first shock pressure in the range 100 - 200 GPa (1 - 2 Mbar)
- Choice of NIF shock pressure is constrained by implosion design considerations: Need to keep shell on low adiabat
- But this is a dangerous range of pressure --- shock at 100 GPa will maximize microstructure-seeded velocity fluctuation
- So in initial Omega experiments we will test effect of shock pressure in seeding instability
- We will vary strength of initial shock from 100 GPa to 200 GPa, thus varying amplitude of seed, while keeping slab instability (growth factor) constant

**Greatest velocity fluctuation is induced for ~100 GPa shock, where elastic precursor disappears on *a* axis, but not *c* axis.**

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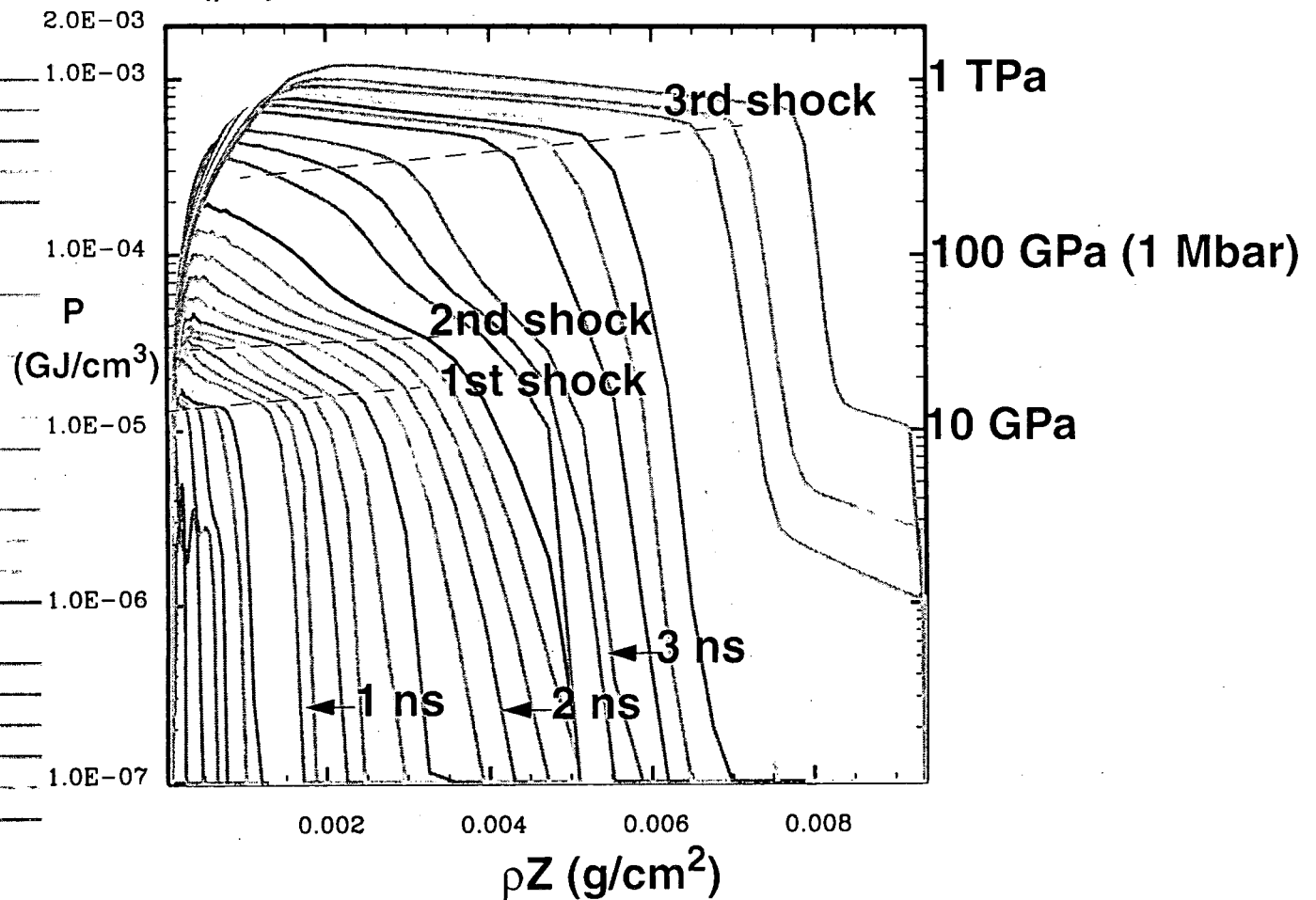




**Current proposed Omega drive pulse generates first and second shock in range 10 - 30 GPa in first 20  $\mu\text{m}$  of slab.**

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* PROBLEM 7 PLOT 1
* PROBLEM NAME OF59
* K= 2 L1= 2 L2= 132
* FILE OF59DB
* TIME 1.0073034E-02
* CYCLE 115
* X VAR: RHLP
* Y VAR: PR
* PROBLEM 7 PLOT 2
* TIME 2.0096275E-02
* PROBLEM 7 PLOT 3
* TIME 3.1168213E-02
* PROBLEM 7 PLOT 4
* TIME 3.9977327E-02
* PROBLEM 7 PLOT 5
* TIME 5.2495305E-02
* PROBLEM 7 PLOT 6
* TIME 6.0693699E-02
* PROBLEM 7 PLOT 7
* TIME 7.0026768E-02
* PROBLEM 7 PLOT 8
* TIME 8.0077605E-02
* PROBLEM 7 PLOT 9
* TIME 9.0613037E-02
* PROBLEM 7 PLOT 10
* TIME 0.1023247
* PROBLEM 7 PLOT 11
* TIME 0.1151696
* PROBLEM 7 PLOT 12
* TIME 0.1297130
* PROBLEM 7 PLOT 13
* TIME 0.1451607
* PROBLEM 7 PLOT 14
* TIME 0.1616510
* PROBLEM 7 PLOT 15
* TIME 0.1779491
* PROBLEM 7 PLOT 16
* TIME 0.1928843
* PROBLEM 7 PLOT 17
* TIME 0.2070062
* PROBLEM 7 PLOT 18
* TIME 0.2213471
* PROBLEM 7 PLOT 19
* TIME 0.2371576
* PROBLEM 7 PLOT 20
* TIME 0.2527102
* PROBLEM 7 PLOT 21
* TIME 0.2680932
* PROBLEM 7 PLOT 22
* TIME 0.2827811
* PROBLEM 7 PLOT 23
* TIME 0.2970087
* PROBLEM 7 PLOT 24
* TIME 0.3256248
* PROBLEM 7 PLOT 25
* TIME 0.3395295
* PROBLEM 7 PLOT 26
* TIME 0.3512569
* PROBLEM 7 PLOT 27
* TIME 0.3610037
* PROBLEM 7 PLOT 28
* TIME 0.3701731
* PROBLEM 7 PLOT 29
```

**P( $\rho Z$ ) for zones in slab at fixed times**



**Other sources of perturbation include engineering features associated with fabrication: fill tubes, plugs, joints, etc.**

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- **We plan to investigate effect of such “initially nonlinear” perturbations in Omega experiments**
  - *Specification for allowable fabrication features will be based on modeling and experiments*
- **Still other sources of perturbation are associated with microstructure, although not with crystal anisotropy**
  - *Oxygen contaminant may accumulate at grain boundaries*
  - *Copper dopant may be distributed nonuniformly*

## Future work:

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- Design second pulse shape, giving first shock strong enough to melt Be/Cu grains
- Ensure that pulse shapes can actually be produced at Omega
- Make theoretical predictions of magnitude of velocity fluctuations in Be or Be/Cu that seed ART instability
- Choose concentration of copper dopant